

Advent of Diamond Electronics:

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**Motivation
Challenges
Progress**

Comparative Material Properties (for Electronic Applications)

Material	E _g eV	ε	μ cm ² /Vs	E _c 10 ⁶ V/cm	vsat 10 ⁷ cm/s	λ W/cmK	n _i cm ⁻³
Si	1.1	11.8	1350	0.3	1	1.5	1.50E+10
Ge	0.66	16	3900	0.1	0.5	0.6	2.40E+13
GaAs	1.4	12.8	8500	0.4	2	0.5	1.80E+06
GaP	2.3	11.1	350	1.3	1.4	0.8	7.70E-01
GaN	3.39	9	900	3.3	2.5	1.3	1.90E-10
3C-SiC	2.2	9.6	900	1.2	2	4.5	6.90E+00
4H-SiC - a axis	3.26	10	720	2	2	4.5	8.20E-09
4H-SiC - c axis	3.26	10	650	2	2	4.5	8.20E-09
6H-SiC - a axis	2.86	9.7	370	2.4	2	4.5	2.40E-05
6H-SiC - c axis	2.86	9.7	50	2.4	2	4.5	2.40E-05
Diamond	5.45	5.5	1900	5.6	2.7	20	1.60E-27
CVD Diamond	5.45	5.5	3800	10	2.8	22	1.60E-27
AlN	5.61	8.7	1100	1.2	1.8	1.7	

Silicon Carbide Benefits and Advantages for Power Electronics Circuits and Systems

AHMED ELASSER, MEMBER, IEEE, AND T. PAUL CHOW, SENIOR MEMBER, IEEE

PROCEEDINGS OF THE IEEE, VOL. 90, NO. 6, JUNE 2002

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Revised Figures of Merit

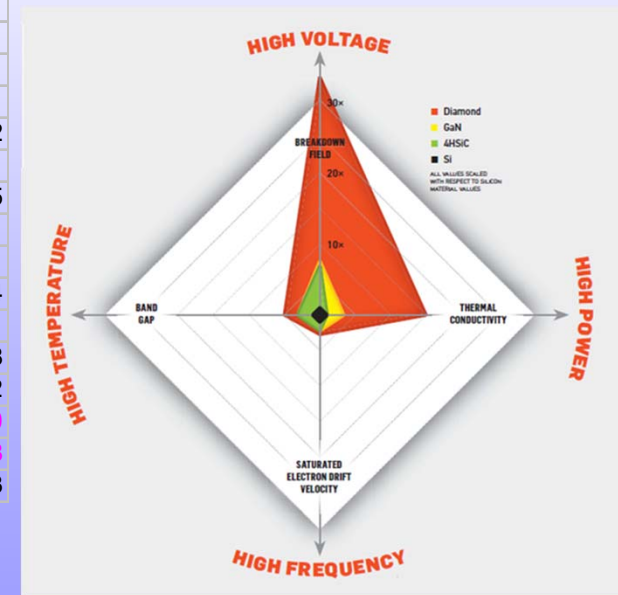
Material	basic limit on the device performance			Power Density & transistor switching speed			Thermal properties, heat sink		Low-frequency Resistive Losses		High-frequency switching losses
	JM	MJM	KM	Qf1	Qf2	BM(Qf3)	BHFM				
	$(E_c v_{sat} \pi^{-1})^2$	$\lambda \cdot JM$	$\lambda \sqrt{v_{sat} \epsilon^{-1}}$	$\lambda \sigma_A$	$\lambda \sigma_A E_c$	$\sigma_A = \epsilon \mu E_c^3$	μE_c^2				
Si	1	1	1	1	1	1	1				
Ge	0.03	0.01	0.24	0.06	0.02	0	0.32				
GaAs	7	2	0	5	7	16	11				
GaP	37	20	1	11	46	20	5				
GaN	756	655	2	587	6454	677	81				
3C-SiC	64	192	5	104	417	35	11				
4H-SiC - a axis	178	533	5	402	2679	134	24				
4H-SiC - c axis	178	533	5	363	2419	121	21				
6H-SiC - a axis	256	768	5	346	2769	115	18				
6H-SiC - c axis	256	768	5	47	374	16	2				
Diamond	2540	33870	32	56905	1062227	4268	490				
CVD Diamond	8712	127769	36	712868	23762274	48605	3128				
AlN	52	59	2	44	174	38	13				

Silicon Carbide Benefits and Advantages for Power Electronics Circuits and Systems

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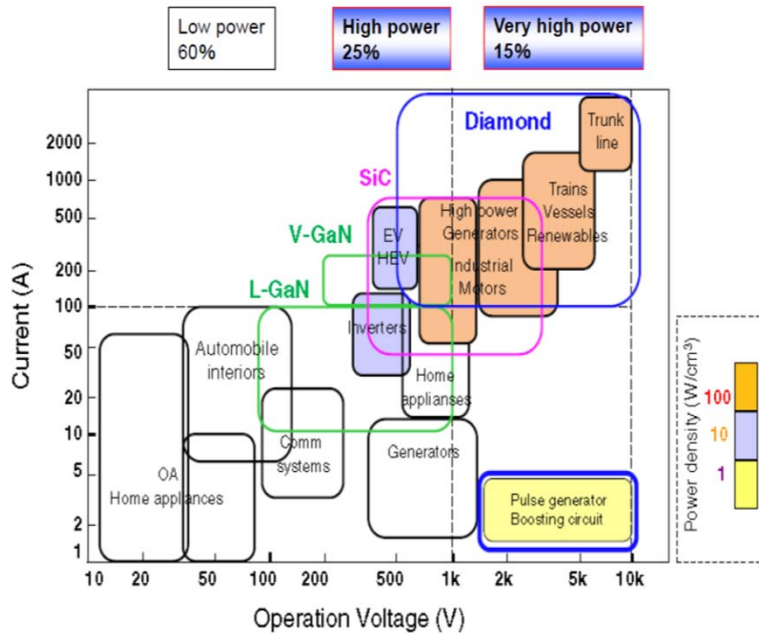
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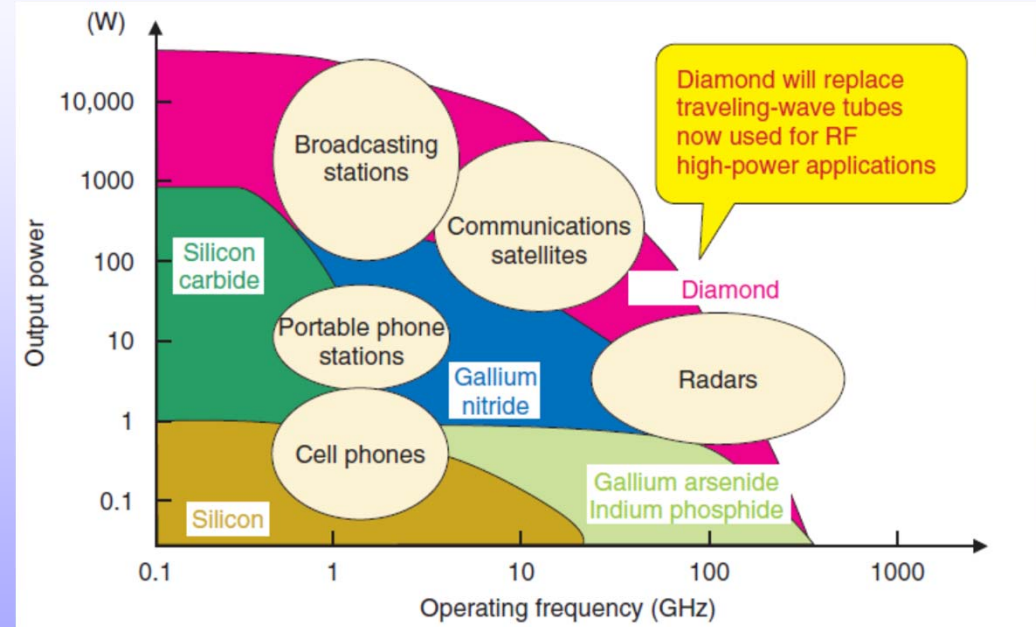
From: www.egr.msu.edu/fraunhofer-ccd/

Probable Applications for Diamond Electronics

Diamond estimated applications



DRM65(2016) 168

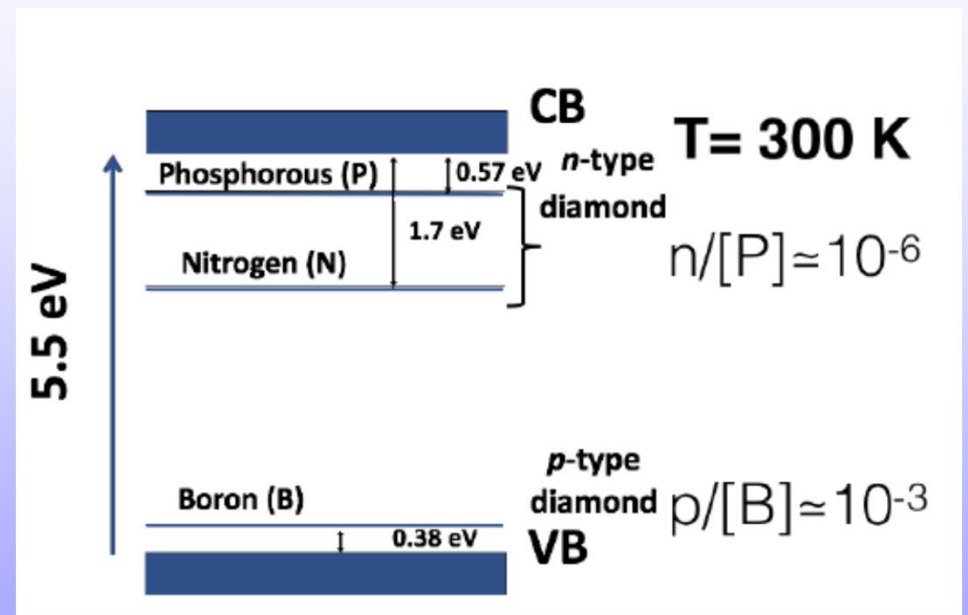


M. Kasu, NTT Tech. Review,

S-i. Shikata – AIST Japan

Requirements for Diamond Electronics

- ◆ Need charge carriers
 - ◆ Electrons in conduction band
 - ◆ Holes in valance band
- ◆ Need High Mobility
- ◆ Need high breakdown strength
- ◆ Need High thermal diffusivity
- ◆ Cost Effective Production



Traditional Issues Limiting Diamond Electronics

◆ Impurities and Defect structure

- ◆ Natural Diamonds – Lack on Uniformity: “they are like snowflakes, no 2 are identical”

- ◆ N impurities
- ◆ dislocations

◆ HPHT

- ◆ N, Metal solvent impurities
- ◆ Dislocations

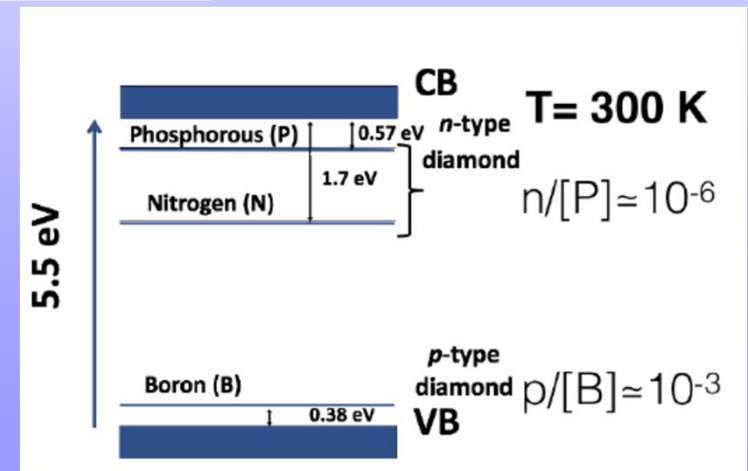
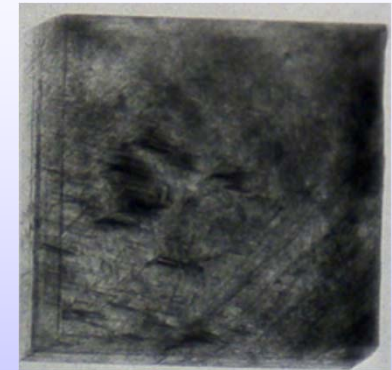
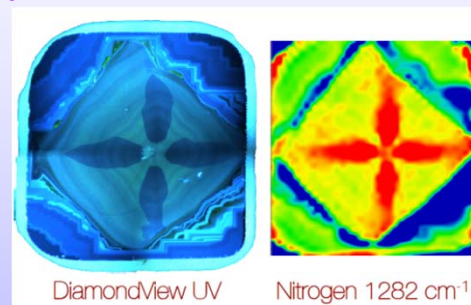
◆ CVD

- ◆ Dependent on substrate seed crystallography
- ◆ Good control of purity

◆ Doping – electrically active impurities

- ◆ Few atoms are small enough to fit into diamond lattice
 - ◆ B, N, H, Si, Ni, ...
 - ◆ Electrically active dopants are energetically removed from the conduction or valance bands

◆ Size and Availability for mass market



Diamond as a Semiconductor

Creating Charge Carriers in Diamond

◆ Bulk Doping

- ◆ P-type - Boron 0.37 eV
- ◆ N-type - Phosphorous 0.6 eV

◆ Surface Doping

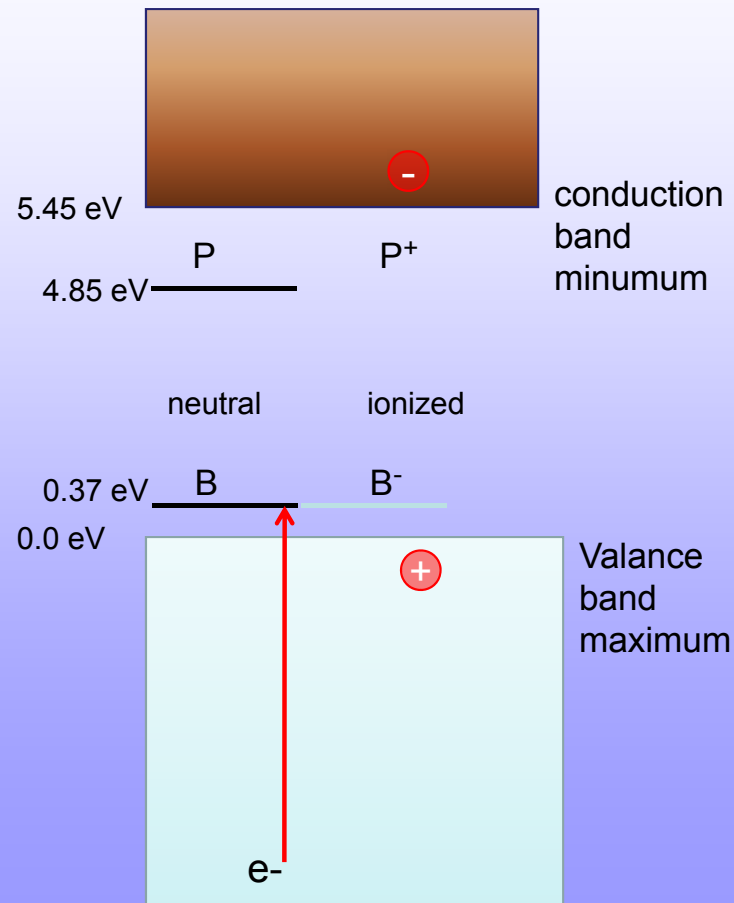
- ◆ P-type - surface adsorbate creates 2D hole gas in subsurface region

◆ “Delta Doping”

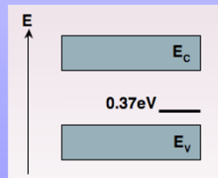
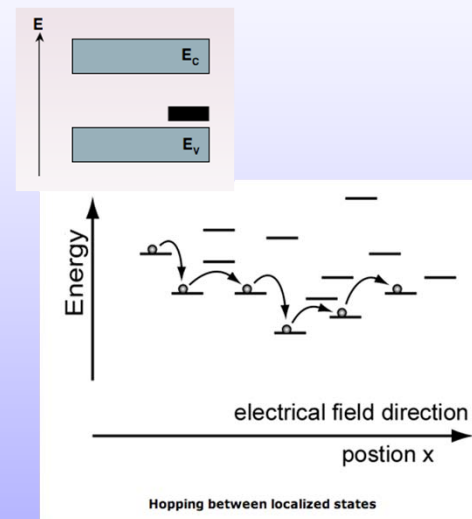
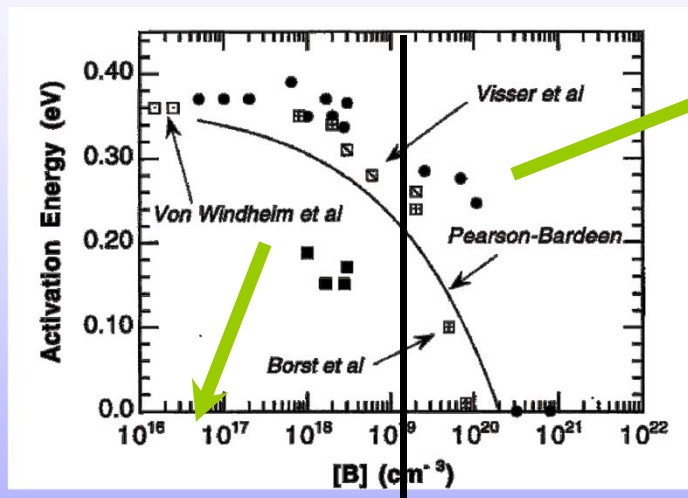
◆ Vacuum Electronics

◆ Photo carrier generation

◆ Quantum Electronics and Sensors



Doping diamond with Boron (bulk)



Impurity band formation, hopping conduction, low mobility
Requires little thermal activation
Conductivity at room temperature

Discrete acceptor state, band conduction, high mobility
Requires high activation energy
Little conductivity at room temperature

Activation energy graph from
Borst & Weiss, Phys. Stat. Sol., A154, 423 (1996)

Carrier Concentration

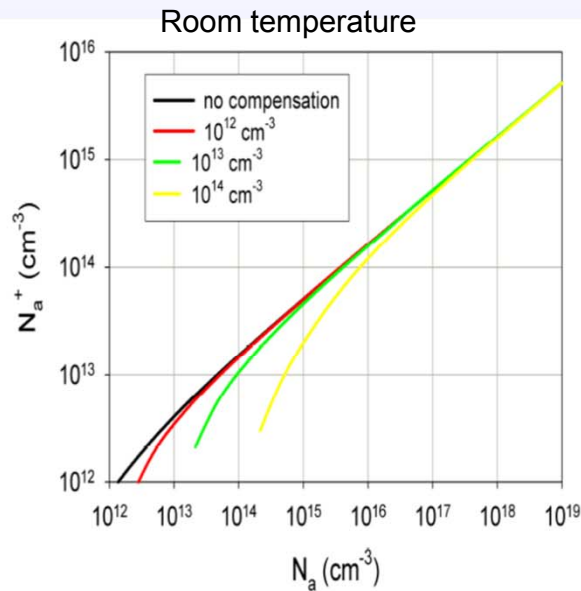


FIGURE 1. Hole concentration vs. dopant concentration of boron doped diamond at room temperature for different concentrations of compensating donors. At higher boron concentrations higher than 10^{19} cm^{-3} the activation increases rapidly to full activation and conduction becomes metallic-type.

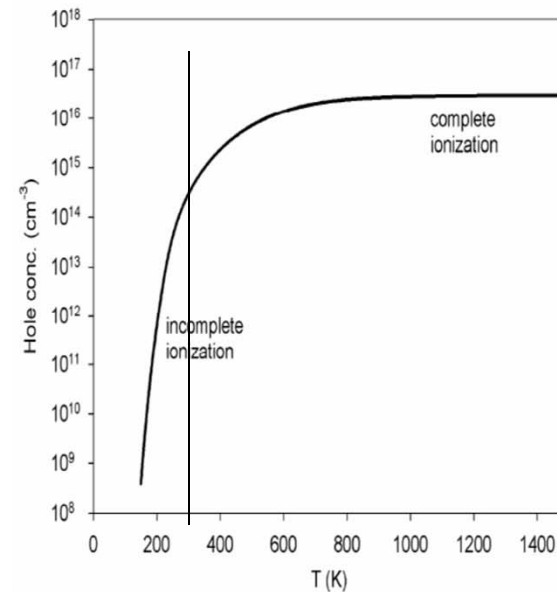


FIGURE 2. Hole concentration vs. temperature of boron doped diamond with dopant concentration $N_A = 4 \cdot 10^{16} \text{ cm}^{-3}$ and compensating donor concentration $N_D = 2 \cdot 10^{14} \text{ cm}^{-3}$.

Diamond Electronic Devices CP1292, 2010 *Wide Bandgap Cubic Semiconductors: From Growth to Devices*
edited by G. Ferro and P. Siffert

© 2010 American Institute of Physics 978-0-7354-0847-0/10/\$30.00

J. Isberg^a

Hall Mobility

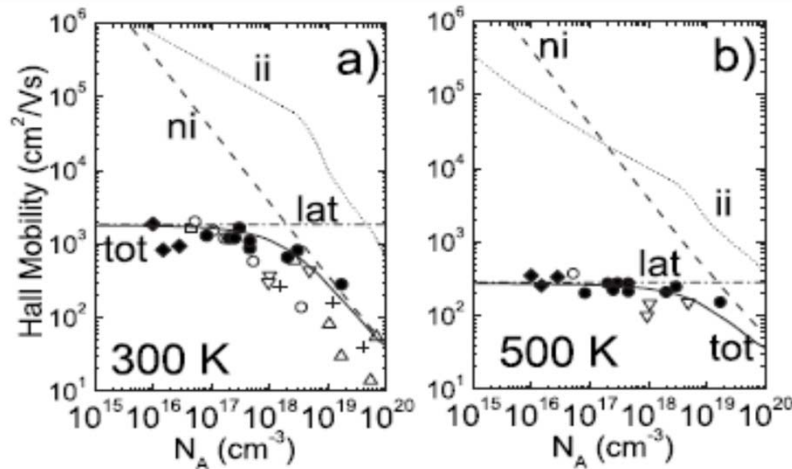


FIG. 4. Hole Hall mobility (σ_{RH}) as a function of doping level in homoepitaxial diamond: (a) at 300 and (b) at 500 K. The Symbols are experimental data ($\blacklozenge$ —Ref. 15, $\bullet$ —Ref. 11, $\square$ —Ref. 12, $\circ$ —Ref. 13, ∇ —Ref. 14, $\triangle$ —Ref. 7, and $+$ —Ref. 9). The theoretical contribution of various scattering modes are illustrated by dashed and dotted line for lattice (lat=ac+op), dotted line for ii mode and dashed line for ni mode, calculated from Eqs. (1), (4), (6), and (7).

Above MIT ($N_a > 4 \times 10^{20} \text{ cm}^{-3}$)
Mobility = 3-5 cm^2/Vsec

PHYSICAL REVIEW B 81, 205203 (2010)

Hall hole mobility in boron-doped homoepitaxial diamond

J. Pernot,* P. N. Volpe, F. Omnès, and P. Muret

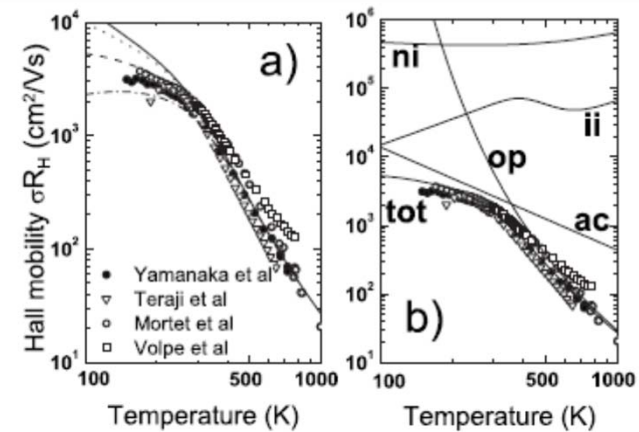


FIG. 1. Temperature dependence of the Hall mobility (σ_{RH}). Experimental data are illustrated by symbols: full circles—Yamanaka *et al.* (Ref. 8), triangles—Teraji *et al.* (Ref. 12), open circles—Mortet *et al.* (Ref. 13), and open squares Volpe *et al.* (Ref. 15). (a) The lines show the calculated mobility with $N_A = 10^{16} \text{ cm}^{-3}$ and different compensation densities: $N_D=0$ —solid line, $N_D=10^{14} \text{ cm}^{-3}$ —dotted line, $N_D=10^{15} \text{ cm}^{-3}$ —dashed line, and $N_D=5 \times 10^{15} \text{ cm}^{-3}$ —dashed-dotted line. (b) The solid lines show the theoretical contribution of various scattering modes and the total mobility (tot) with $N_A=10^{16} \text{ cm}^{-3}$ and $N_D=10^{15} \text{ cm}^{-3}$. The scattering mechanisms are ii, ni, ac, and op, calculated respectively from Eqs. (1), (4), (6), and (7).

Delta Doping of Diamond with Boron

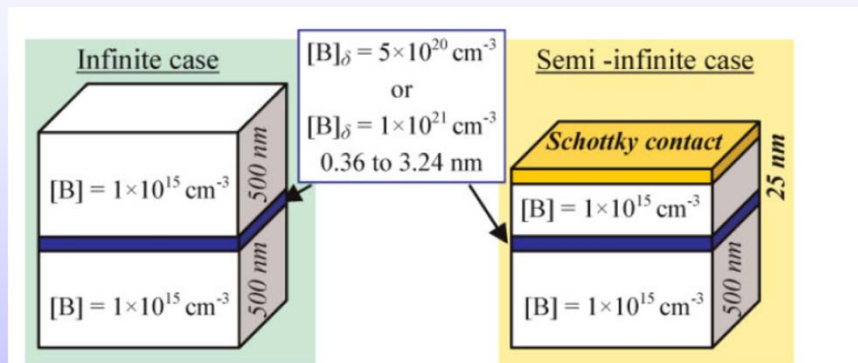


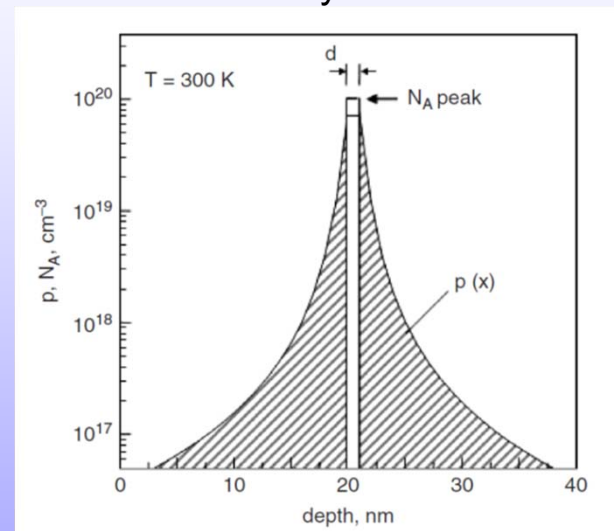
Figure 1 (online color at: www.pss-a.com) Delta-boron-doped diamond stacks simulated in this work.

Simulations of carrier confinement in boron δ -doped diamond devices

Alexandre Fiori*, Julien Pernot**, Etienne Gheeraert, and Etienne Bustarret

Phys. Status Solidi A **207**, No. 9, 2084–2087 (2010) / DOI 10.1002/pssa.201000062

Carrier density vs distance



E. Kohn and A. Denisenko

p. 319 in *"CVD Diamond for Electronic Devices and Sensors"*

Ed. R.S. Sussmann

Delocalization of Holes

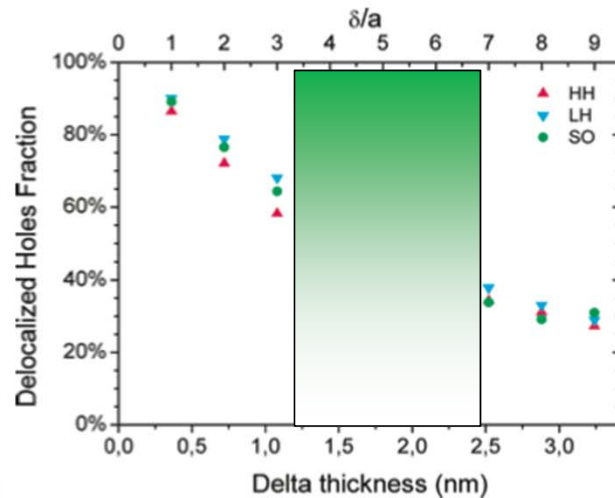


Figure 4 (online color at: www.pss-a.com) Delocalized holes fraction in an infinite diamond structure with a boron concentration of $1 \times 10^{21} \text{ cm}^{-3}$.

seven atomic bilayers, three energy levels were occupied by holes (HH, LH, and SO). These energy levels correspond to

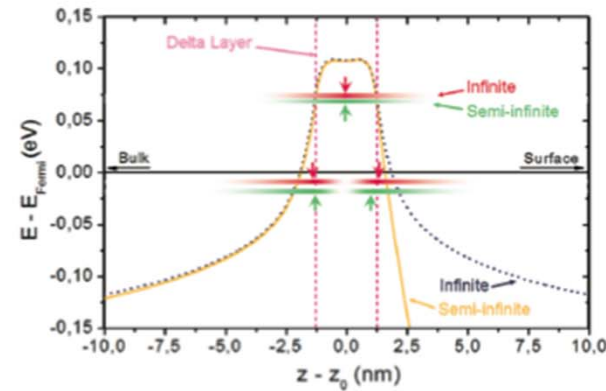


Figure 5 (online color at: www.pss-a.com) Heavy holes probability density is colored in pink for $\delta = 2.52 \text{ nm}$ in an infinite diamond structure (valence band potential is drawn in blue) and in green in the semi-infinite structure (valence band potential is drawn in gold). Peaks of heavy holes probability density are localized by wide arrows. The vertical position of the energy level indicates its energy, the horizontal position its spatial extension, and the color contrast its probability density. The Fermi level is drawn in black. Only the ground state and first excited state are represented to simplify the drawing.

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Delta-doping: Mobility of previous experiments

- Other materials- enhancement:
 - GaAs, GaN – mobility 2D structures even higher than in 3D
- Diamond (other research groups 2000-now): **no enhancement**

French group: [G. Chicot et al. Appl. Phys. Lett. 101, 162101 (2012)]
[G. Chicot et al. J.Appl.Phys., 116, 083702(2014)] ***Mobility = 3.6 (0.8) cm²/Vsec***

E6+Germany: [H. El-Hajj et al, DRM, 17, 2008, 409] ***Mobility = 0.01 to 0.1 cm²/Vsec***
[J. Scharpf et al. Physica Status Solidi a-Applications and Materials
Science 2013, 210, 2028]

Element 6: [R. S. Balmer et al. Journal of Applied Physics 2013, 113.] ***Mobility = 3.8 cm²/Vsec***

OUR STRATEGY

- ◆ High Quality, low dislocation density HPHT (100) substrates
- ◆ Polish to 0.1 to 0.3 nm Ra
- ◆ Remove polishing damage
- ◆ Design ultra fast gas switching laminar flow CVD reactor
- ◆ Grow smooth epilayers VERY slowly, ca. 30 to 70 nm/hr
- ◆ Chemically getter residual boron contamination during growth of intrinsic layers

Substrate preparation for atomically smooth surface

1. Polishing of substrate, single crystal (100) diamond ($R_a < 0.3 \text{ nm}$)

80 x80 microns

$S_a = 0.15 \text{ nm}$

$S_q = 0.19 \text{ nm}$

2. ICP(Inductively coupled plasma)-RIE etching

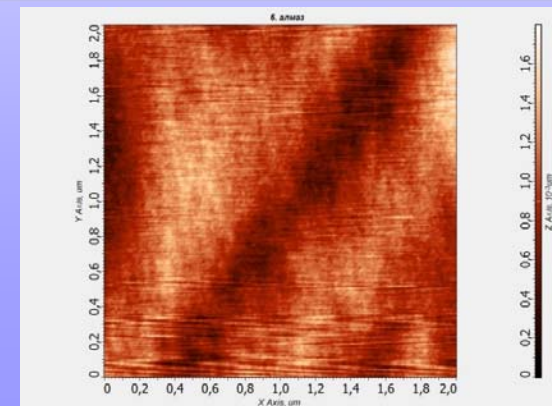
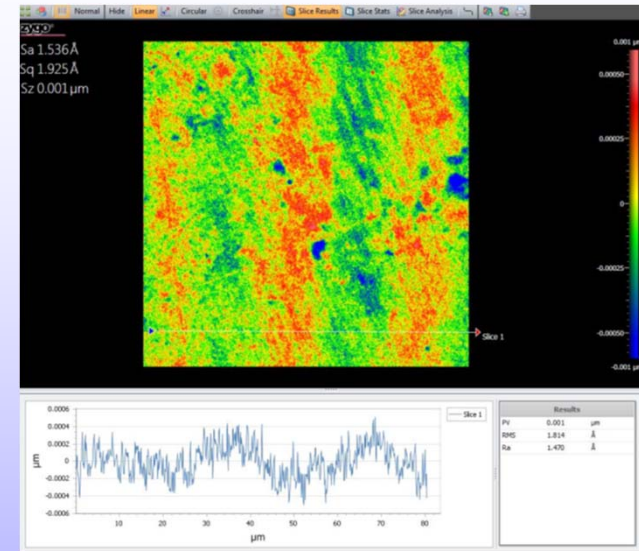
ICP etching of diamond is highly homogeneous, does not increase surface roughness and eliminate polishing damages.

[A.B. Muchnikov et al., Phys. Status Solidi A, 2015]

3. Surface after CVD growth, $R_a < 0.25 \text{ nm}$

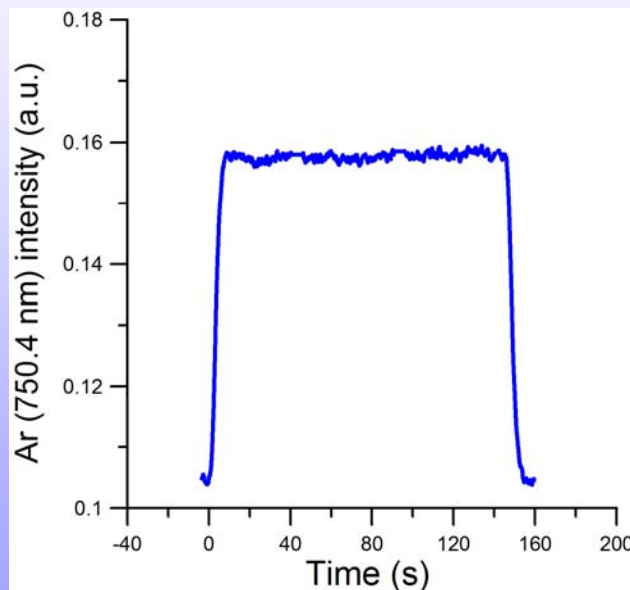
2x2 microns
AFM

Zygo Opt. Interferometer



Main features of delta doping reactor

1. Rapid gas switching allows achieving fast changing of gas mixtures containing B_2H_6 and without B_2H_6
2. Laminar gas flow in reactor – vortices and stagnant regions are absent

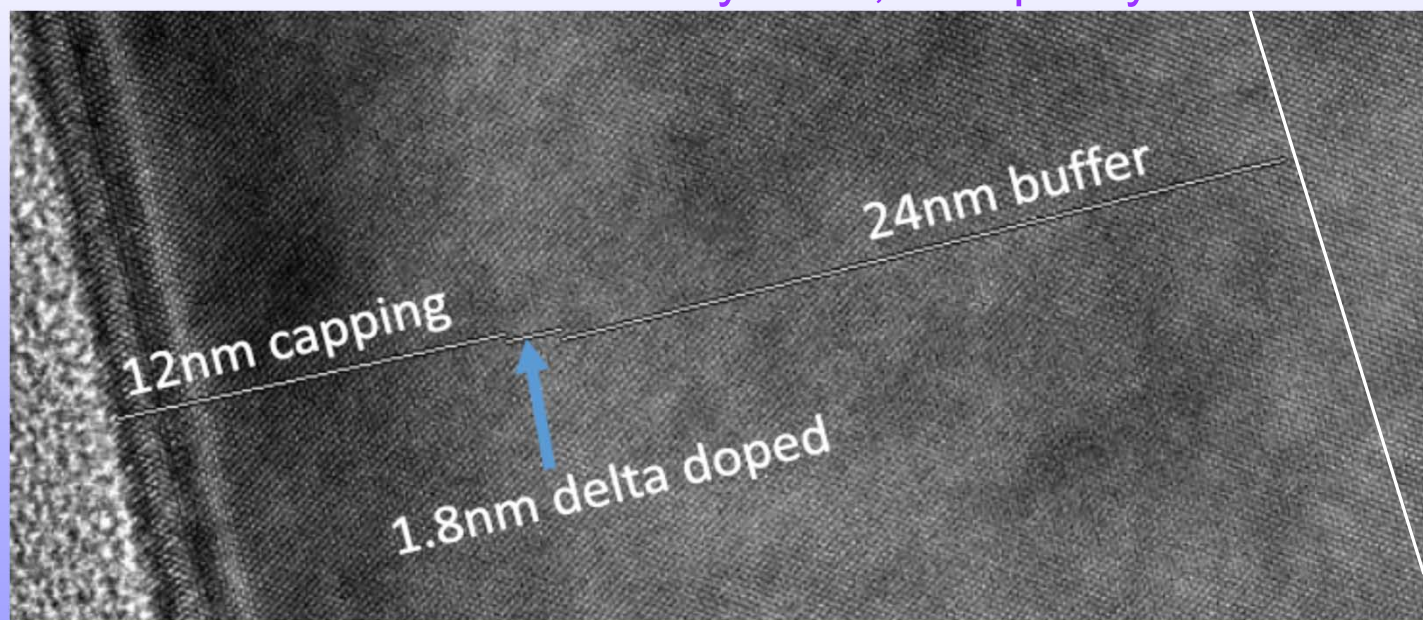


Intensity of Ar (added in diagnostic purposes) line on time after changing gas mixture from H_2-CH_4 to H_2-CH_4-Ar and back. Total gas flow 950 sccm.

3. Axial symmetric resonant mode – symmetric discharge
4. Slow growth of diamond 40-100 nm/h
low methane content $<0.1\%$, low pressure 40 Torr

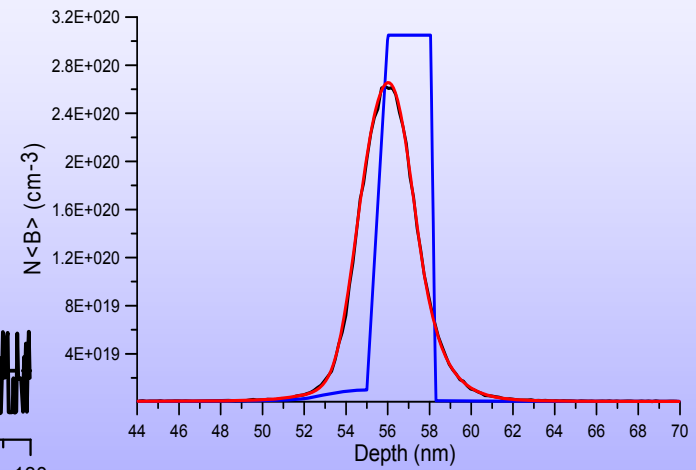
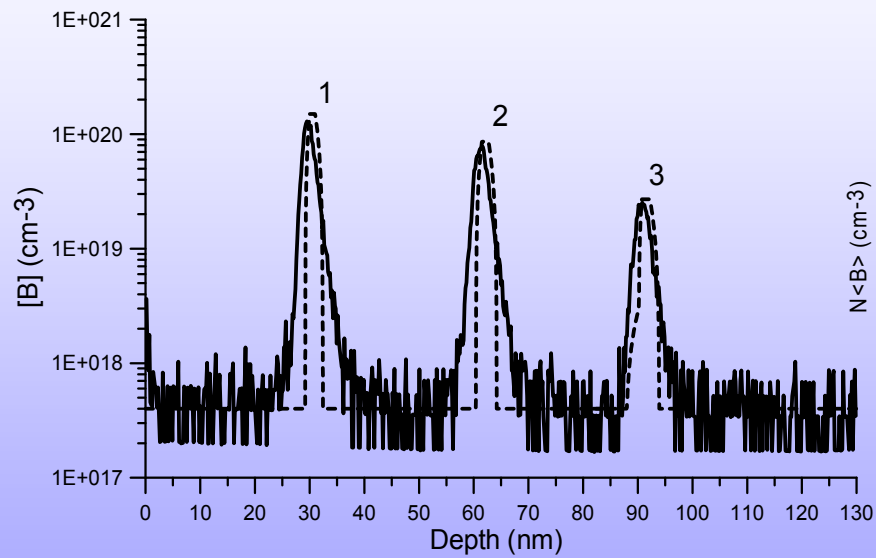
Interfaces are barely seen, full epitaxy.

substrate



CVD deposition

SIMS sputter profiling

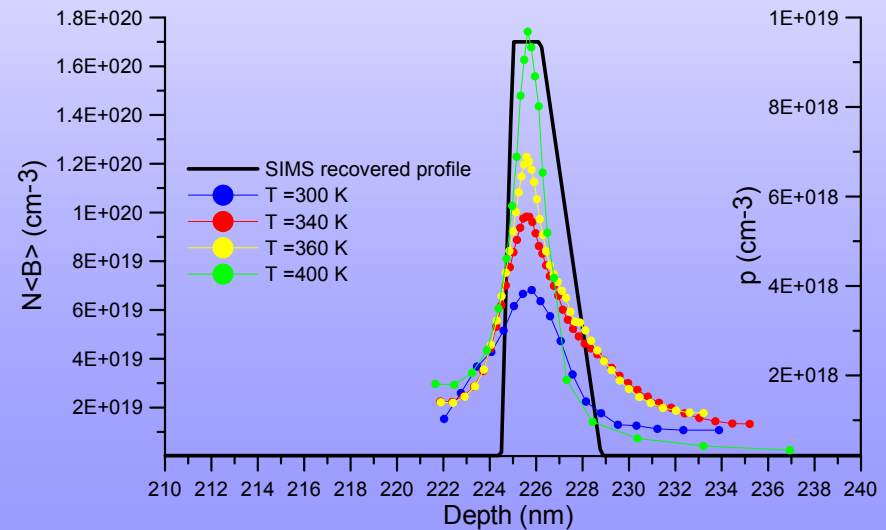
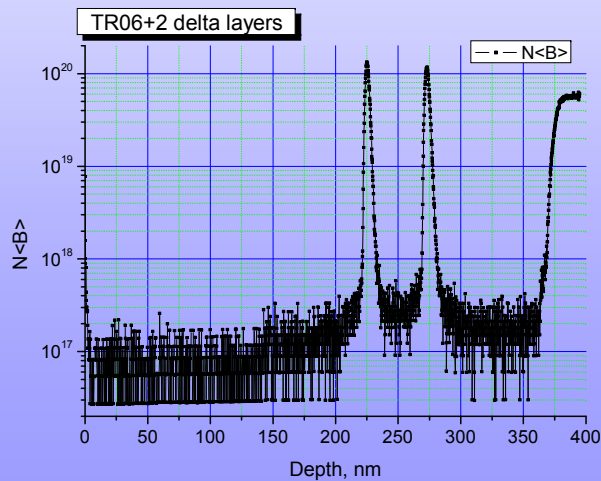
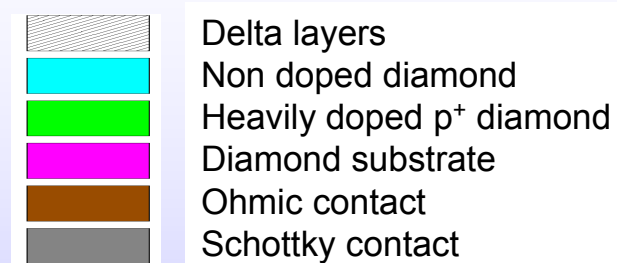
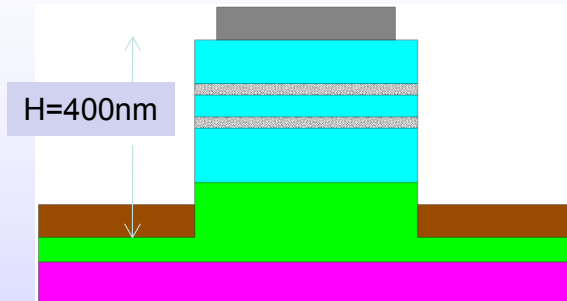


Deconvolute SIMS Instrument function:

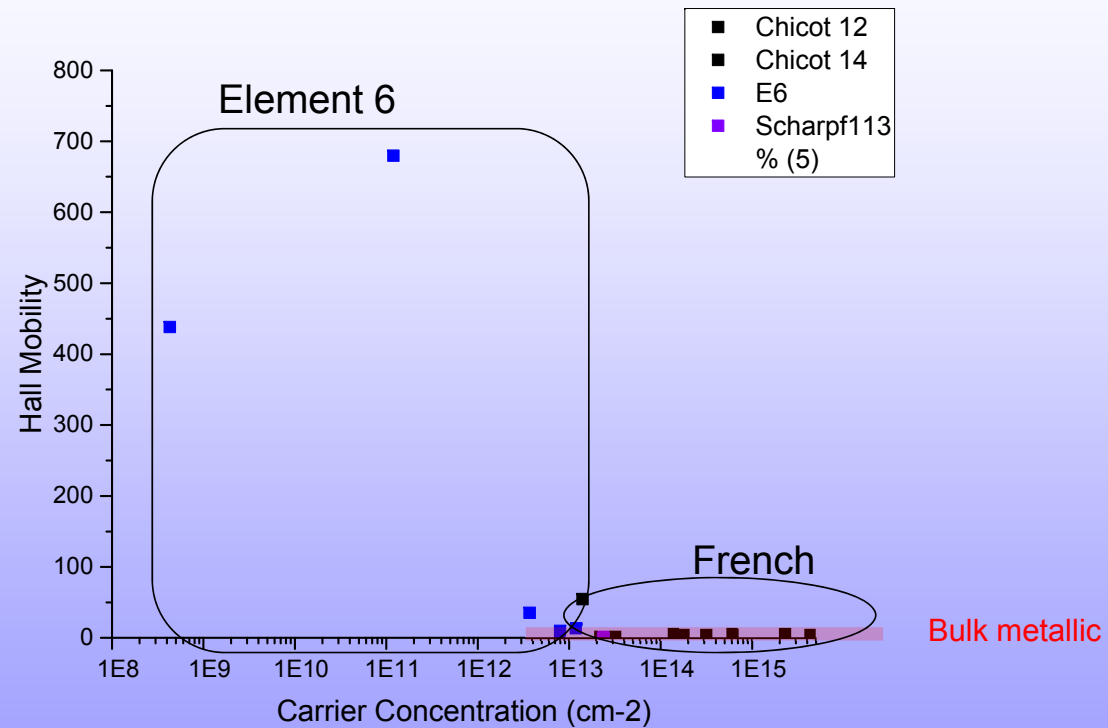
S. Hofmann, Y. Liu, J. Y. Wang, J. Kovac,
Applied Surface Science 2014, 314, 942.

CV of vertical mesa with delta layers

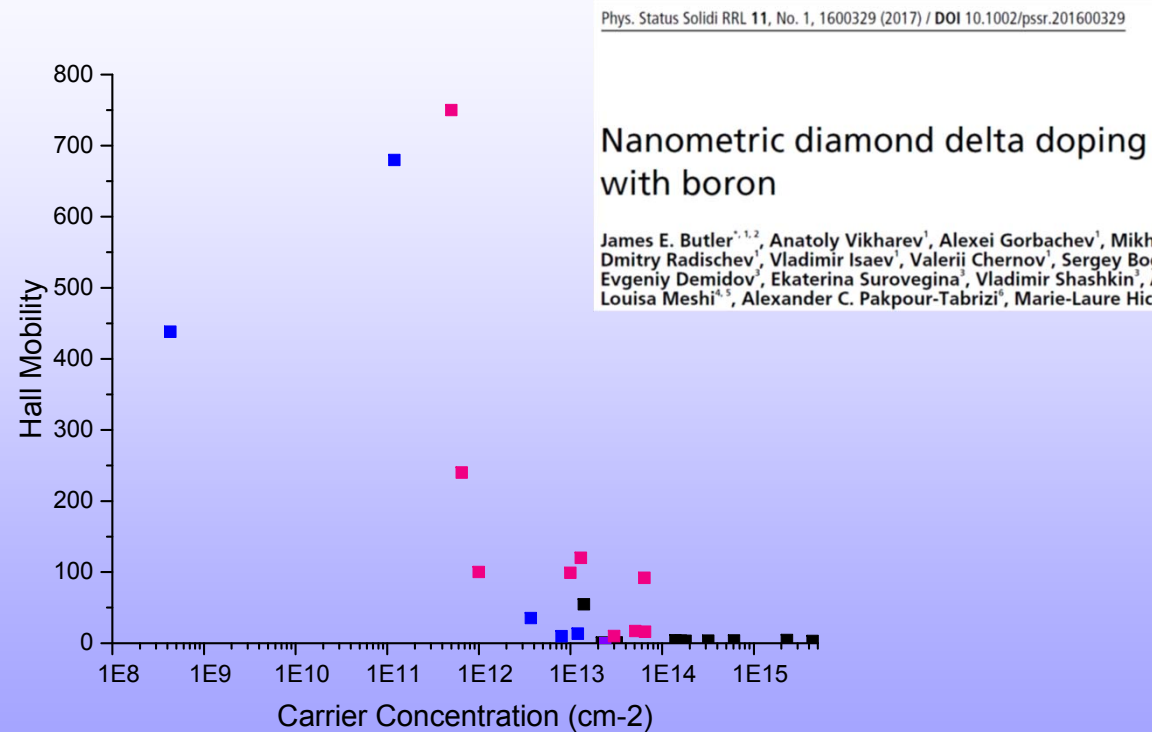
Formation of the mesa structure with contact to the bottom P⁺ layer



Previous Hall Measurements (*v_{dp}* and Hall bar)

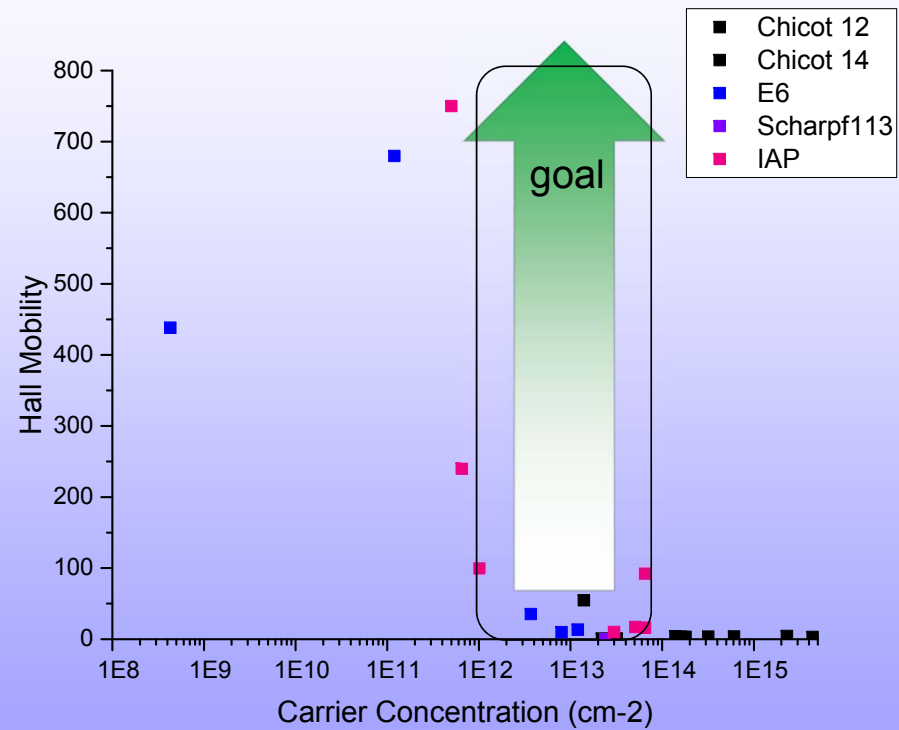


Initial Hall Measurements (vdp config.)



Initial Hall Measurements

(vdp config.)



Summary

- ◆ Diamond Electronics is becoming a Reality!
- ◆ Why?
 - ◆ Available High Quality Substrates (HPHT)
 - ◆ High Purity Epitaxial Layer Growth (CVD)
 - ◆ Reliable Doping Methods
 - ◆ Traditional substitutional doping with B, P
 - ◆ H terminated surface conduction
 - ◆ Boron Delta Doping
- ◆ Thank You for Your Attention.

Acknowledgements

- ◆ Naval Research Laboratory – Office of Naval Research (1975-2010)
- ◆ Russian Science Foundation ‘Megagrant’ (2013-2017) Act 220 of Duma
- ◆ New Diamond Technology – St. Petersburg Russia